Journal of

NANOSTRUCTURES



Investigating the effects of post-heat treatment temperatures on the structure of prepared nanorod by hydrothermal method

Masood Hamadanian^{a, b, *}, Hani Sayahi^b, Ali Reza Zolfagharici^b

Article history:

Received 24/12/2011

Accepted 23/2/2012

Published online 1/3/2012

Keywords:

nanorod

photoluminescence (PL) spectroscopy

Dye-sensitized solar cells (DSSCs) Hydrothermal method

Titanum dioxide (TiO₂)

*Corresponding author:

E-mail address:

Hamadani@kashanu.ac.ir Phone: +98 361 5912382

Fax: +98 361 5912397

Abstract

In this study, the TiO_2 nanorods were synthesized from P25 TiO_2 nanoparticles by hydrothermal method in 10 M NaOH solution. The effects of annealing temperatures on produced nanorods were investigated by scanning electron microscopy (SEM) X-ray diffraction (XRD) and photoluminescence (PL) spectroscopy. Diameter growth and changes of surface oxygen defects of synthesized nanorods are studied with increasing annealing temperatures.

2012 JNS All rights reserved

1. Introduction

In recent years, the nanostructured metal oxide semiconductors have widely employed due to their promising applications in electronic, optical, chemical and mechanical properties [1-3]. Among the various metal oxide semiconductors, titanium

dioxide (TiO₂) has attracted great attention because of its potential to apply in environmental cleaning [4, 5], photocatalysis [6, 7], sensors [8, 9], lithium ion batteries [10] and solar cells [11, 12]. Therefore, many researches recently have been done to improve and recognize the structures and properties of

^aInstitute of Nano Science and Nano Technology, University of Kashan, Kashan, I.R. Iran.

^bDepartment of Physical Chemistry, Faculty of Chemistry, University of Kashan, I.R. Iran.

^cDepartment of Physical Chemistry, Chemistry and Chemical Engineering Research Center of Iran, Tehran, I.R. Iran.

TiO₂nanostructures. One of the most attractive research areas was synthesized and investigated onedimensional (1D) TiO₂ nanostructures, because of their potential applications in a variety of novel devices [1, 13-15]. Various TiO2-derived 1D nanostructures (nanotube, nanorod, nanofiber) have been synthesized by various techniques, such as template method [16, 17], anodic oxidation [18], and hydrothermal method [19, 20]. Since, the last technique was introduced in 1988 by Kasuga et al. [19] there are much research has been targeted towards an improved understanding of the mechanism of 1D TiO2 formation in alkali solution and to establish the crystal structure of the nanostructers [21-24].

Titania 1D nanostructures could be synthesized by a reaction of NaOH and TiO_2 nanoparticles with anatase or rutile structure. It has been generally recognized that the obtained nanstructures are built in titanate compounds such as $M_2Ti_3O_7$ or M_xH_2 . $_xTi_3O_7$ (M = Na or K) [25, 26]. Normally raw TiO_2 nanoparticles, such as a commercial anatase powder P25 (Degussa Co., Germany, a mixture of anatase and rutile; its size is less than 30 nm), are usually used as the raw powder reacted with NaOH or KOH [27–29].

In this paper, we have investigated the effect of post heat treatment on microstructure and morphology of synthesized TiO₂ nanorod by hydrothermal method. We have also presented detailed scanning electron microscopy (SEM) studies on morphology and structures of titanate nanorod. In addition, the effect of post heat treatment on surface oxygen vacancies and defects

of produced TiO₂ nanorod have been investigated by photoluminescence (PL) spectroscopy [30, 31].

2 Experimental

The TiO₂ source used for producing the nanorod was a commercially TiO₂ powder (P25, Degussa AG, Germany), which consists of 20% rutile and 80% anatase. The primary particle size was about 21 nm. For preparing TiO₂ nanorod 1.5 g of TiO₂ powder was added to 100 ml of 10M NaOH solution. Next, the TiO₂ powder was dispersed by ultrasonic bath (15 min) and vigorous stirring (30min) respectively. The solution was placed in a PTFE (Teflon) lined autoclave and heated for 48 hours at 150 °C. After the hydrothermal treatment, the resulting powder was cooled to room temperature, and then the filtered precipitate washed with water and 0.1M HCl solution until the washing solution achieved pH $6 \sim 7$. At the end, the washed powder was dried in oven at 100 °C for 24 hours.

The obtained powder was sintered at 300 and 500 °C to study the post heat treatment effects on the TiO₂ nanorods morphology and structure.

The morphologies of prepared TiO_2 nanorods were characterized by scanning electron microscopy (SEM, Hitachi S-4160). The crystal structure was analyzed from XRD patterns (using $Cu-K_{\alpha}$ radiation, BRUKER, D8 advance). The adsorption/desorption isotherms of these samples were measured by N₂-adsorption at 77 K (BEL, Japan Inc) using Barrett-Joyner-Halenda (BJH) and Brunauer-Emmett-Teller (BET) calculations to charcterize the porosity and surface area of the prepared nanorod, respectively. In addition, the post-

heat treatments effects on the synthesized nanorods were characterized by photoluminescence spectroscopy (Jasco, FP 6500).

3. Results and discussion

Figure 1a shows the morphological images of TiO₂ nanoparticles as precursor for hydrothermal nanorod formation. After hydrothermal reaction for 48 h and the post hear treatment for 24 h, TiO₂ nanoparticles converted to 1D structure nanorod with about 75 nm and few hundred nm in length (figure 1b).

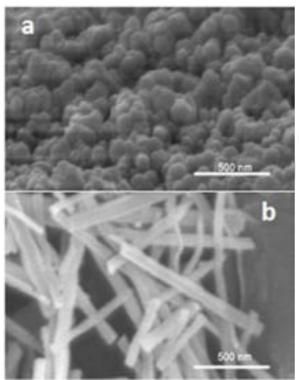


Figure 1. SEM images of (a) P25 nanoparticles as precursor (b) the synthesized nanorod after annealing at 100 °C.

The BET specific surface area ($S_{\rm BET}$), pore volume (V_P), and pore diameter for prepared nanorod (figure 1b) was about 60.06 (m²/g), 0.27

(cm³/g) and 18.54 nm, respectively. Adsorption/desorption isotherms of synthesized sample also indicated that TiO2-derived titanate nanorod have mesoporous structures.

Figure 2 shows the SEM images of prepared nanorod after annealing at 300 (2a) and 500 °C. The SEM results indicated that nonorods diameters were increased in results of increasing the post heat treatment at 300 °C, but we didn't observe change in diameter at annealing at 500 °C. The nanorods diameters were become about 125 nm after post heat treatment at temperatures over 300 °C. This may be due to decreasing of surface charge and nanorods approaching together. Another point, the nanorods were more compact with relation to increase of annealing temperatures [32, 33].

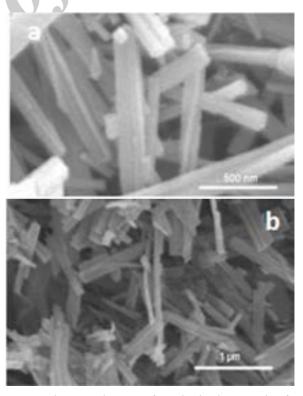


Figure 2.The SEM images of synthesized nanorods after annealin at a) 300 b) 500 °C.

As shown in figure 3, the X-ray diffraction (XRD) pattern of precursor nanoparticles (P25) is compared with produced nanorods after annealing at 100, 300 and 500 °C. As expected, the peaks in the diffraction patterns for the nanorods broadened due to dimensional confinement and the new peaks were become evident in $2\theta=28$, 46 and 58, because of 1D TiO₂ nanostructures[34, 35]. The rutile structures peaks of TiO2 nanoparticles were eliminated after hydrothermal processes and all synthesized sample have an anatase structures $(2\theta=25, 29, 48 \text{ and } 63)$. We didn't anneal the prepared nanorods over 500 °C due to possibility of phase conversion from anatase to rutile. The anatase phase of TiO₂ structures is more important than rutile for using in photocalytic and solar system because rutile is the most stable phase while anatase possesses superior optoelectronic and photocatalytic properties [36, 37]. We used the synthesized 1D anatase nanorods in dye-sensitized solar cell system and the obtained data will be present elsewhere.

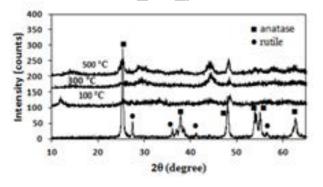


Figure 3. XRD patterns of prepared nanorods at different annealing temperatures comparing to $P25 \ TiO_2$ nanoparticles.

We also investigated the effect of post heat treatment surface structures on the photoluminescence spectroscopy (PL). Figure 4 shows the PL spectra of synthesized nanorods annealed at different temperatures with excitation wavelength at 300 nm to understand the changes of the nanorods surfaces structures. The excitonic PL intensity of nanorods samples decreases as the calcination temperature increases, which is due to the increase in size of nanorods with increasing temperature. And, calcination this demonstrates that the excitonic PL mainly results from surface oxygen vacancies and defects [30, 31]. Therefore, increasing oxidation of surface of TiO₂ with relation to enhance of heat treatment temperatures cause decrease of surface oxygen vacancies and defects and overall decrease of PL intensity. On the other hand, reducing the surface defects on TiO₂ nanostructures improve the photocatalytic and optoelectronic properties.

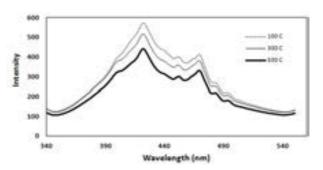


Figure 4. The PL spectra of synthesized nanorods annealed at different temperature with excitation wavelength of 300 nm

4. Conclusion

In this study, the effects of post heat treatment temperatures were in detail investigated by SEM, XRD and PL spectroscopy. The obtained results were shown that the increasing of annealing temperature until 300 °C causes increase of thicknesses of prepared nanorods by hydrothermal method. Increase of annealing temperature over 300 °C has no effect on the size of produced nanorods, but due to reducing of the surface charge causes the more compact products. In addition, the surface defects of synthesized 1D TiO₂ were decreased because of increase of annealing temperatures, but we should note that at higher temperatures (over 500 °C) the TiO₂ phase may be converted from anatase to rutile.

References

- [1] X. Gan, X. Gao, J. Qiu, P. He, X. Li, X. Xiao, Cryst. Growth Des., 12 (2012)289.
- [2] T. Tatsuma, S. Saitoh, P. Ngaotrakanwiwat, Y. Ohko, A. Fujishima, Langmuir, 18 (2002) 7777.
- [3] Y. Tong, Y. Liu, L. Dong, D. Zhao, J. Zhang, Y. Lu,D. Shen, X. Fan, J. Phys. Chem. B, 110 (2006) 20263.
- [4] M. Andersson, L. Osterlund, S. Ljungstrom, A. Palmqvist, J. Phys.Chem. B 414 (2002) 338.
- [5] D. Robert, J. Lede, Chimie, Soleil, energie. Environmental, Special issue, (2000) 228.
- [6] H. Tada, F. Suzuki, S. Ito, T. Akita, K. Tanaka, T. Kawahara, H. Kobayashi, J. Phys. Chem. B, 106 (2002) 8714.
- [7] T. Kubo, M. Takeuchi, M. Matsuoka, M. Anpo, M. Nakahira, A.Catal. Lett. 130 (2009) 28.
- [8] C. Guo, F. Hu, C. Ming Li, P. K. Shen, Biosensors and Bioelectronics, 24 (2008) 825.
- [9] A. Rothschild, A. Levakov, Y. Shapira, N. Ashkenasy, Y. Komem, Surf. Sci, 456 (2003) 532.
- [10] Y. Wang, M. Wu, W.F. Zhang, Electrochimica Acta, 53 (2008) 7863.

- [11] B.O. Reagen, M. Gratzel, Nature, 353 (1991) 373.
- [12] M. Hamadanian, H. Sayahi, A.R. Zolfaghari, JNS 1 (2012) 139.
- [13] A. Nakahira, T. Kubo, C. Numako, Inorg. Chem, 49 (2010) 5845.
- [14] S. Iijima, Nature, 354 (1991)56.
- [15] F. Shao, J. Sun, L. Gao, So. Yang, J. Luo, J. Phys. Chem. C. 115 (2011) 1819.
- [16] P. Hoyer, Langmiur, 12 (1996) 1411.
- [17] H. Imai, Y. Takei, K. Shimizu, M. Matsuda, H. Hirashima, J. Mater. Chem, 9 (1999) 2971.
- [18] D. Gong, C. A. Grimes, O. K. Varghese, W. Hu, R.S. Singh, Z. Chen, E. C. Dickey, J. Mater. Res, 16 (2001) 3331.
- [19] T. Kasuga, M. Hirashima, A. Hoson, T. Sekino, K. Niihara, Langmuir, 14 (1998) 3160.
- [20] C. C. Tsai, H. Teng, Chem. Mater, 18 (2006) 367.
- [21] B. D. Yao, Y. F. Chan, X. Y. Zhang, W. F. Zhang, Z.Y. Yang, N. Wang, Appl. Phys. Lett., 82 (2003) 281.
- [22] Q. Chen, G. H. Du, S. L. M. Zhang Peng, Acta Crystallogr., Sect. B, 58 (2002) 587.
- [23] T. Kasuga, M. Hiramatsu, A. Hoson, T. Sekino, K. Niihara, Adv. Mater., 15 (1999) 1307.
- [24] D. V. Bavykin, V. N. Parmon, A. A. Lapkin, F. C. Walsh, j. Mater. Chem, 14 (2004) 3370.
- [25] Y. B. Mao, S. S. Wong, J. Am. Chem. Soc. 128 (2006) 8217.
- [26] B. X. Wang, X. P. Zhao, Adv. Funct. Mater. 15 (2005) 1815.
- [27] J. B. Yin, X. P. Zhao, Nanotechnology, 17 (2006) 192.
- [28] C.C. Tsai, H. Teng, Chem. Mater. 16 (2004) 4352.
- [29] L. Qian, F. Teng, Z. S. Jin, Z. J. Zhang, T. Zhang, Y.B. Hou, S. Y. Yang, X. R. Xu, J. Phys. Chem. B, 108 (2004) 13928.

[30] J. Liqiang, Q. Yichun, W. Baiqi, L. Shudan, J. Baojiang, Y. Libin, F. Wei, F. Honggang, S, Jiazhong, Solar Energy Materials & Solar Cells, 90 (2006) 1773.

[31] L. Q. Jing, B. F. Xin, D. J. Wang, F. L. Yuan, H. G. Fu, J. Z. Sun, Chem. J. Chin. Univ., 26 (2005) 111.

[32] H Hu, C. Deng, X. Huang, Materials Chemistry and Physics, 121 (2010) 364.

[33] X. Q. Zhao, C. R. Kim, J. Y. Lee, C. M. Shin, J. H. Heo, J. Y. Leem, H. Ryu, J. H. Chang, H. C. Lee, C. S. Son, B. C. Shin, W. J. Lee, W. G. Jung, S. T. Tan, J. L. Zhao, X. W. Sun, Applied Surface Science, 255 (2009) 5861.

[34] G. Armstrong, A. Robert Armstrong, J. Canales, P. G. Bruce, Chem. Commun., (2005) 2454.

[35] J. Jiu, S. Isoda, F.Wang, M. Adachi, J. Phys. Chem. B,110 (2006) 2087.

[36] C. Wang, Z. X. Deng, Y. Li, Inorg. Chem.,40 (2001) 5210.

[37] J. Jen Wu, C. C. Yu, J. Phys. Chem. B, 108 (2004) 3377.



Investigating the effects of post-heat treatment temperatures on the structure of prepared nanorod by hydrothermal method

Masood Hamadanian^{a, b,} *, Hani Sayahi^b, Ali Reza Zolfagharici, ^b

^aInstitute of Nano Science and Nano Technology, University of Kashan, Kashan, I.R. Iran.

^bDepartment of Physical Chemistry, Faculty of Chemistry, University of Kashan, I.R. Iran.

^cDepartment of Physical Chemistry, Chemistry and Chemical Engineering Research Center of Iran, Tehran, I.RIran

بررسی اثرات دماهای عملیات حرارتی روی نانو میله تهیه شده به وسیله روش هیدروترمال هیدروترمال چکیده:

در این مطالعه، نانو میله های TiO₂ از نانو ذرات P25-TiO₂ با روش هیدروترمال در محلول TiO₂ از نانو فرات مطالعه، نانو میله های تولیدی به وسیله ی پراش اشعه ایکس، میکروسکوپ تهیه شده اند. اثرات دماهای گداختگی روی نانو میله های تولیدی به وسیله ی پراش اشعه ایکس، میکروسکوپ الکترونی روبشی و فوتولومینسانس مورد بررسی قرار گرفته اند. رشد قطری و تغییرات نقص اکسیژن سطحی نانو میله های سنتزی با افزایش دمای گداختگی مورد مطالعه قرار گرفته است.